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Effect of Dicumyl Peroxide Concentration on the Polymerization Kinetics of a Polysilazane System

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The effect of dicumyl peroxide (DCPO) concentration on the polymerization kinetics of a polysilazane system has been investigated through DSC analyses. DCPO acts as polymerization initiator by activating the polyaddition of vinyl groups. The overall heat of reaction is found to be a linear function of the DCPO concentration above 1.5 wt%, due to the addition of the heat of fully initiated polymerization and the heat of DCPO decomposition. Below 1.5 wt% of DCPO, the polymerisation of polysilazane is not fully initiated and only part of the reaction occurs. The resulting overall heat of reaction is therefore lower than that provided by the linear law valid for higher concentrations. A two-reaction kinetic model built on this interpretation leads to a satisfactory representation of the DSC analyses performed for each DCPO concentration between 0.1 and 20 wt%.

INTRODUCTION

Discovered in 1885 [1], polyorganosilazanes encompass a wide variety of silicon carbo-nitriles precursors utilized since the 1950s to prepare ceramic materials [2, 3] from a single component [4]. Processed at the liquid state, they are particularly well suited for the preparation of Ceramic Matrix Composites (CMC) by Polymer Impregnation and Pyrolysis (PIP) [5–7]. Indeed, this multi-step process involves first the injection of the polymer into a fibrous preform and then the ceramization of the matrix through the pyrolysis of the precursor. The present work is a contribution to the optimization of CMC processed by PIP.

A first step of optimization was performed thanks to several works [8–13] aimed to reduce the volatilization of silazane oligomers in polyorganosilazanes formulations. Higher molecular weights were thereby achieved, together with denser polymers in the intermediate material and denser ceramics in the final product. The general molecular structure of polysilazanes is shown in Fig. 1, in an oversimplified way. By changing R_1 and R_2 groups, it is possible to adjust the electronic, optical and rheological properties of the material [14].

More recently, the studies focused on the mechanical and physical in-use properties of polysilazane-derived ceramics [15–18] or on the reactions involved in the polymer-to-ceramic

transformation [19]. However the literature related to the processing properties of the precursors is much less detailed, except properties related to pyrolysis step [5, 6]. A few papers investigate the properties of the intermediate polymer. In particular, viscosity is a key feature to design and optimize the PIP process, and is obviously strongly dependent on polymerization. The viscoelastic behavior of polysilazane derived preceramic polymers was investigated by Flores et al. [20, 21] and D'Elia et al. [22]. From an industrial point of view, optimizing the process means to reduce the processing time without affecting the material quality, and inter alia requires the crosslinking kinetics to be known, minimized and controlled. Crosslinking of polysilazanes is usually performed by heating at 100–250°C [23, 24], or alternatively by UV radiation [25, 26].

Several reactions are involved in the crosslinking mechanism [27] (cf. section “Polysilazane Crosslinking Chemistry”), and some of them can be promoted using the suitable initiator or catalyst. For instance, Flores et al. [20] used tetra-*n*-butylammoniumfluoride to select a crosslinking mechanism involving dehydrogenation with Si-H and N-H groups to control the glass transition temperature, viscoelastic behavior and thermal stability of the polymer in order to optimize the processing of non-oxide ceramic fibers.

For processing CMC by PIP process, other mechanisms involving reactions without gaseous products (for instance polyaddition of vinyl groups) are more suitable to limit the matrix porosity.

Faster kinetics at lower temperature are made possible by adding a radical initiator, such as organic peroxide, most of the time dicumyl peroxide (DCPO) [24, 28–31], which is able to initiate polyaddition of vinyl groups [32]. The DCPO concentration recommended to reduce the crosslinking temperature ranges between 0.01 and 5 wt% [33–35], but a comprehensive study on the effect of peroxide concentration on polymerization kinetics remains mandatory in the framework of material and process optimization.

DCPO is widely used as crosslinking agent in the polymer industry. Owing to a very exothermal heat of reaction (between 666 and 868 J·g⁻¹ [36–40]), DCPO decomposition kinetics has been thoroughly studied in many conditions. In most cases, a pseudo-first order kinetic is reported, with reaction order between 3/4 and 1 [36–40]. However DCPO decomposition kinetics was proved to be sensitive to environment [41]: in the absence of oxygen, DCPO decomposition follows a pseudo-first order kinetics model, whereas an autocatalytic model is more adequate in the presence of oxygen.

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FIG. 1. General molecular structure of a polysilazane system.

The scope of the present paper is to analyze the changes in the polymerization kinetics of a polysilazane system with different DCPO concentrations, with the aim to identify the optimal value and to account for DCPO concentration in a crosslinking kinetic model, on the basis of the autocatalytic model identified in [22] for a DCPO concentration of 1wt%.

EXPERIMENTAL

The commercial polysilazane under study is the Clariant® KiON® Ceraset® PSZ 20. Stored in a dark, cool (4°C) and dry place, the shelf-life of this pale yellow liquid is 6 months [23]. Its density ranges between 1 and 1.1 g·cm⁻³, and its viscosity between 180 and 750 mPa·s at 20°C. Once crosslinked between 100 and 250°C, the liquid is converted to a solid polymer consisting of an alternated Si-N backbone with ~80% methyl (CH₃) and ~20% vinyl (CH=CH₂) side groups as shown in Fig. 2 [24].

According to the manufacturer, the polymer solidifies after 2 h at 100°C, 1 h at 150°C, 40 minutes at 200°C and 20 minutes at 250°C in air. In an inert gas atmosphere, curing takes a longer time. Curing can be accelerated by addition of dicumyl peroxide (DCPO). The dicumyl peroxide (≥98%) used in this work is produced by Sigma Aldrich® and is stored in a dry, dark and cool place (4°C). The chemical formula (C₆H₅C(CH₃)₂OOC-(CH₃)₂C₆H₅) of DCPO, is presented in Fig. 3 that shows two benzene rings, four methyl groups (CH₃) and the classical peroxide group (O-O) [42]. Its molar mass is 270.37 g·mol⁻¹.

Polysilazane and DCPO are mixed for 1 h at 293 K (20°C) using a magnetic stirrer, up to attain the complete dissolution of the peroxide and then stored at 277 K (4°C) for 12 h before testing in order to eliminate the air bubbles formed during mixing. Several dicumyl peroxide concentrations were investigated between 0.1 wt% and 20 wt%.

A series of preliminary tests was conducted using a Thermo-Gravimetric-Analysis (TGA) coupled with a Differential Scanning Calorimetry (DSC) under dynamic conditions, with a heating rate of 10 K·min⁻¹, using a TGDSC 111 of SETARAM. The study of dynamic cure kinetics change with peroxide concentration is then performed by Differential Scanning Calorimetry. All DSC tests are conducted on the compounds under

FIG. 2. Chemical structure of PSZ20 (Ceraset®) polymer provided by Clariant® [24].

FIG. 3. Dicumyl peroxide chemical formula: C₆H₅C(CH₃)₂OOC-(CH₃)₂C₆H₅ [42].

dynamic conditions, with a heating rate of 2 K·min⁻¹. DSC tests on DCPO alone were conducted under dynamic conditions, with heating rates of 0.5, 2, 5 and 10 K·min⁻¹. A DSC-Q200 of TA Instruments was used to measure the specific heat of reaction (ΔH_{tot}) of the polymer and the specific heat of decomposition of the DCPO (ΔH_{DCPO}). The initial mass of the samples ranges between 10 and 20 mg and all tests were performed under continuous N₂ flow (50 ml·min⁻¹) using a non-hermetic crucible. Temperature and enthalpy were calibrated with pure indium under the same conditions of heating rate, gas flow and crucible as used in the tests.

RESULTS AND DISCUSSION

DCPO Decomposition

Figure 4 depicts the results of DSC scans performed on DCPO samples at 0.5, 2, 5 and 10 K·min⁻¹. The heat of decomposition, ΔH_{DCPO} , ranges between 1,002 J·g⁻¹ (5 K·min⁻¹) and

FIG. 4. Dynamic DSC analyses of DCPO decomposition at heating rates of 0.5, 2, 5 and 10 K·min⁻¹ and simulated results using Eq. 1 with (a), the set of parameters A and (b), the set of parameters B, (cf. Table 1).

TABLE 1. DCPO decomposition kinetics parameters identified from the present results compared with literature data [36–41].

		ΔH_{DCPO} (J·g ⁻¹)	E_i (kJ·mol ⁻¹)	k_i (s ⁻¹)	n_i
Wu et al. [36]		741.15–750.52	125.35–125.39	$3.06\text{--}3.12 \times 10^{12}$	0.86–0.90
Wu et al. [37]		666–813	132–139	2.15×10^{13} to 1.44×10^{14}	0.85–1.00
Ben Talouba et al. [38]		855.2–867.8	135 ± 2	$1.18 \pm 0.017 \times 10^{10}$	1.01 ± 0.01
Chen et al. [39]		774–837	161–203	NA	3/4
Duh et al. [40]		810.5	129.7	1.26×10^{13}	1
Di Somma et al. [41]		NA	129.9 ± 0.2	$6.5 \pm 0.83 \times 10^{12}$	1
Present work		1051 ± 37			
Set A	1 st peak ($i = 1$)	$\Delta H_1 = 147$	105	5.0×10^{10}	0.5
$\zeta = 0.87$	2 nd peak ($i = 2$)	$\Delta H_2 = 904$	139	1.64×10^{14}	1
Set B	1 st peak ($i = 1$)	$\Delta H_1 = 221$	110	2.25×10^{11}	0.58
$\zeta = 0.79$	2 nd peak ($i = 2$)	$\Delta H_2 = 831$	153.1	8.81×10^{15}	1

1,091 J·g⁻¹ (0.5 K·min⁻¹). The average value, $1,051 \pm 37$ J·g⁻¹, is significantly higher than the results of other studies, between 666 and 868 J·g⁻¹ [36–40]. However, the behavior is quite different with two distinct peaks, whereas literature only reports one single peak. The nature of both peaks was not further investigated in the present study. It is noteworthy that a weak secondary peak has been neglected in [40].

Actually, an n^{th} -order model with coefficients reported in [36, 37, 40, 41] (Table 1) is able to properly describe the principal peak occurring in the present experimental results (second peak occurring at higher temperature in Fig. 4). Starting from the parameters reported in [36, 37, 40, 41] (Table 1) and assuming a pseudo first-order model for the second peak, a first set of parameters, labeled A, has been identified to fit the present experimental data with two uncoupled n^{th} -order model (Eq. 1), each related to a part of the total heat of reaction (ζ for the second peak and $(1-\zeta)$ for the first one). The set of parameters A considers that 87% of ΔH_{DCPO} is due to the second peak. The set of parameters A is within the range of values reported in [36, 37, 40, 41] (Table 1) and provides a heat of reaction for the principal peak, ΔH_2 , of 904 J·g⁻¹, slightly higher than ΔH_{DCPO} values reported in [36–40].

$$\dot{\alpha}_{\text{DCPO}} = k_1 \cdot e^{\frac{-E_1}{RT}} \cdot (1-\zeta-\alpha_1)^{n_1} + k_2 \cdot e^{\frac{-E_2}{RT}} \cdot (\zeta-\alpha_2)^{n_2} \quad (1)$$

Assuming that the first peak does not affect the specific heat flow at the second one, the Kissinger method [43] was applied to identify the energy of activation. A value of 153.1 kJ·mol⁻¹ was found and lead to a second set of parameters, labeled B, still assuming a pseudo first-order model for the principal peak. This method provides the best fit to experimental data and a heat of reaction ΔH_2 of 831 J·g⁻¹ (79% of ΔH_{DCPO}) was found for the principal peak, within the range of ΔH_{DCPO} reported in [36–40]. The value of 153.1 kJ·mol⁻¹ for the energy of activation is significantly higher than those reported in [36–38, 40], between 125 and 139 kJ·mol⁻¹. However Lv et al. [39] found a value of energy of activation between 161 and 203 kJ·mol⁻¹ using the Friedman isoconversional method.

Polysilazane Crosslinking Chemistry

The crosslinking reactions of polysilazanes involve mainly the groups -CH=CH₂ (vinyl), Si-H and N-H. The main reactions, described hereinafter, are polyaddition of vinyl groups,

radical reactions of methyl and vinyl groups, hydrosilylation, dehydrogenations and transamination. These reactions take place at different temperature ranges. According to Lavedrine et al. [27], the crosslinking starts at about 120°C with the hydrosilylation reaction (1), followed by dehydrogenation of Si-H with N-H (2). Transamination (3) takes place at approximately the same time, followed by polyaddition of vinyl groups (4), and dehydrogenation with two Si-H groups (5). The radical reactions of methyl and vinyl groups (6) occur at higher temperature, between 200 and 300°C [44].

According to the ATG-DSC analysis conducted on the PSZ20 resin alone, no significant reaction is observed up to 200°C (dashed black line in Fig. 5). However according to TGA analysis (dashed grey line in Fig. 5) a mass loss of 2.5% is observed at 200°C, which is attributed to dehydrogenation of Si-H with N-H groups. Above 280°C, an exothermal reaction starts,

FIG. 5. TGA-DSC analyses of the PSZ20 resin alone (dashed lines) and with addition of 1 wt% of DCPO (continuous lines) between 50 and 300°C, at a heating rate of 10 K·min⁻¹.

with a decrease of mass, which corresponds to the reaction of transamination. The sudden mass loss is produced by the higher mass of the ammonia product (NH₃).

The polyaddition of vinyl groups can be activated at low temperature by the addition of a free radical generator as DCPO. According to the literature [16, 45], this reaction takes place for a temperature between 50 and 150°C for a polymethylvinylsilazane resin with addition of a peroxide cure initiator. After addition of 1wt% of DCPO, vinyl polyaddition is promoted with regard to dehydrogenation. This inversion in the reaction order should reduce the mass loss at 200°C. According to TGA results (continuous grey line in Fig. 5), mass loss is reduced by more than 60% at 200°C after addition of 1wt% of DCPO, confirming literature results. A sharp exothermic peak is obtained around 150°C.

Heat of Reaction

Figure 6 shows the dynamic DSC scans at 2 K·min⁻¹ of the different tested solutions. DCPO concentration (represented by the variable x) ranges from 0.1 wt% to 20 wt%. Two effects are immediately observed with the increase of DCPO concentration: a gradual decrease of the peak temperature (from 155°C for 0.1 wt% to 121°C for 20 wt%), and the occurrence of a second exothermal phenomenon for a concentration higher than 3 wt%.

In Fig. 7 are plotted the DSC scans at 2 K·min⁻¹ of the compounds with x values of 1, 5, 10 and 20 wt%, superimposed with the DSC scan of DCPO at 2 K·min⁻¹ multiplied by the corresponding x value. A strong correlation is observed between the second peak of the compound and the principal peak of DCPO decomposition (almost the same amplitude), whose kinetics turns out to be slightly faster in the compound. Owing to these results, the second peak arising in the compounds for high DCPO concentrations will be attributed to DCPO decomposition in the forthcoming interpretations.

The first exothermal peak observed for the compound is therefore related to PSZ20 crosslinking. ATG-DSC results plotted in Fig. 5 show that this peak is not correlated to a significant mass loss. Reactions with gaseous products as dehydrogenation and transamination may thus be limited in the presence of DCPO. Moreover, the radical reactions of methyl and vinyl groups are expected to occur at higher temperature, between

200 and 300°C, according to [44]. Hydrosilylation and polyaddition of vinyl groups (promoted by DCPO) might be the principal reactions occurring in the crosslinking mechanism in the temperature range investigated.

According to this interpretation the overall heat of reaction (ΔH_{tot}), which represents the time integral of the curves plotted in Fig. 6, can be additively decomposed into two components, related on the one hand to the contribution of PSZ20 crosslinking (initiated by DCPO) and on the other hand to DCPO decomposition. Both components depend on the variable x , i.e. the mass fraction of DCPO. The evolution of the overall heat of reaction is presented in Fig. 8 as a function of DCPO concentration and can be well fitted using a bilinear model (Fig. 8), with coefficients of determination of 0.999 and 0.997 respectively. According to the previous discussion regarding the initiation of vinyl group polyaddition by addition of DCPO, it is important to notice that above a certain value of DCPO concentration, the initiator will simply degrade without interacting with PSZ20, forming radicals that will not combine with vinyl groups. Both terms of the overall heat of reaction are thus proportional to the concentration in the concerned reagent, respectively PSZ20 and DCPO, leading to a linear change of the heat of reaction with x . For low DCPO concentrations (below 1.52 wt%), the overall heat of reaction is lower than expected from this linear law. This is interpreted as a lack of DCPO to interact with all the vinyl groups available in the PSZ20 to complete the crosslinking mechanism related to vinyl groups polyaddition. In any case, to attain a complete degree of polymerization, it will be necessary to heat the system at 200–300°C, where transamination and dehydrogenation reactions will take place, involving Si-H and N-H groups.

Above 1.52 wt% of DCPO, ΔH_{tot} can thus be expressed as a linear law, Eq. 2, by introducing the specific enthalpy of polymerization, ΔH_p , and the specific enthalpy of decomposition, ΔH_d .

$$\Delta H_{tot} = \Delta H_p \cdot (1-x) + \Delta H_d \cdot x \text{ if } x \geq 1.52 \text{ wt\%} \quad (2)$$

By fitting the experimental data with Eq. 2, it is possible to assess the value of both specific heats of reaction. For $x=0$, the intercept of Eq. 2 gives the specific heat of polymerization, Eq. 3, and the specific heat of DCPO decomposition in the compound is obtained for $x=1$, Eq. 4.

FIG. 6. Specific heat flow versus temperature of PSZ20 with several DCPO concentrations during dynamic DSC scans at 2 K·min⁻¹.

FIG. 7. Comparison of dynamic DSC scans at $2 \text{ K} \cdot \text{min}^{-1}$ for several compounds with various DCPO concentrations, x , and for DCPO alone weighted by x : (a) $x = 1 \text{ wt}\%$, (b) $x = 5 \text{ wt}\%$, (c) $x = 10 \text{ wt}\%$, and (d) $x = 20 \text{ wt}\%$.

$$\Delta H_p = 231.87 \text{ J} \cdot \text{g}^{-1} \quad (3)$$

$$\Delta H_d = 964.92 \text{ J} \cdot \text{g}^{-1} \quad (4)$$

As explained above, when DCPO concentration is lower than 1.52 wt%, which represents the cross-point between both straight lines fitting the experimental data (Fig. 8), the reaction of polymerization will not be complete in the temperature range investigated, because radicals generated by DCPO decomposition will not be numerous enough to interact with all the active vinyl groups available in the system. In this low concentration range, an exponential-type function, Eq. 5, is introduced to account for the partial initiator effect in order to obtain a smooth function over the entire DCPO concentration range, instead of the bilinear one of Fig. 8.

$$\Delta H_{tot} = \Delta H_p \cdot \left(1 - A_0 \cdot e^{-\frac{x}{x_0}}\right) \cdot (1-x) + \Delta H_d \cdot x \quad (5)$$

Despite its phenomenological origin, Eq. 5 can be interpreted as an increase in the probability for a vinyl group to interact with the initiator with higher DCPO concentration. $1 - A_0$ would represent the part of all the vinyl groups available in the sample that can be initiated by a single DCPO molecule (when x tends toward zero). The remaining vinyl groups (a part A_0 of the total number) would have more probability to interact with DCPO if the concentration is high, what is described by the exponential-type function. It is noteworthy that, according to this interpretation, the value of A_0 and x_0 would probably depend on the mass of the sample and on the heating rate.

According to these considerations, Eq. 5 expresses the overall heat of reaction over the whole DCPO concentration range up to 20 wt% with a single continuous and smooth function. Both parameters A_0 and x_0 are identified in such a way that

Eq. 5 fits all the experimental data, providing the following values: $A_0 = 0.18$ and $x_0 = 0.009$.

The first term of Eq. 5 describes the transition from a partial polymerization of the system to a full one. The result of this equation for the overall heat of reaction is presented in Fig. 9, where the increase of DCPO concentration initially increases the heat of polymerization (exponential term), and then tends toward a linear decrease related to the decrease of PSZ20 concentration, accounted for by the term $(1-x)$. The heat of decomposition obviously increases linearly with x . It can be checked in Fig. 9 that the experimental data of overall heat of reaction are satisfactorily fitted by Eq. 5 using the previously identified values of ΔH_p , ΔH_d , A_0 , and x_0 .

According to this model it is possible to define the maximum degree of conversion for polymerization reaction, Eq. 6, which

FIG. 8. Overall heat of reaction for the compound as a function of DCPO concentration.

FIG. 9. Overall heat of reaction versus DCPO concentration: experimental values (squares) and results of the model (—) accounting for DCPO decomposition (---) and polymerization of PSZ20 (—), including a partial polymerization for low DCPO concentrations.

will be used in the following section to model the cure kinetics of the system as a function of DCPO concentration.

$$\alpha_{pmax} = 1 - A_0 \cdot e^{-\frac{x}{x_0}} \quad (6)$$

Cure Kinetics Modeling

In a previous work [22], cure kinetics of PSZ20 with 1 wt% of DCPO was modeled using a Kamal-Sourour phenomenological model [46]. The parameters were identified using several isothermal DSC scans, taking into account the initial degree of conversion of the system after an initial heating rate. This model proved to properly describe both isothermal and dynamic scans. However the effect on polymerization of DCPO concentration was not considered, because the DCPO concentration, x , was not included in the model. Moreover DCPO decomposition was not considered.

For these reasons a more complex phenomenological model is necessary to describe the reaction kinetics of this system, taking into account a partial degree of conversion for both reactions of polymerization and decomposition. Reaction kinetics will be modeled using two phenomenological models,

FIG. 10. Arrhenius rate constants for the Kamal-Sourour model (from isothermal DSC analyses reported in [22]).

representing the decomposition of the dicumyl peroxide and the polymerization of the polysilazane.

DCPO decomposition will be modeled using an n^{th} -order model, Eq. 7, where n_d is the reaction order and k_d is the rate constant that follows an Arrhenius temperature dependence.

$$\dot{\alpha}_d = k_d e^{-\frac{E_d}{RT}} \cdot (1 - \alpha_d)^{n_d} \quad (7)$$

This model was already successfully applied to the modeling of DCPO decomposition kinetics [36, 37, 39–41] with a reaction order close to 1 (cf. section “DCPO Decomposition” and Table 1). The activation energy ranges from 118 to 139 kJ·mol⁻¹, the reaction order n ranges from 3/4 to 1 and the decomposition enthalpy ranges from 666 to 868 J·g⁻¹ [36, 37, 39–41]. The variability of these parameters depends on the type of identification algorithm used as well as the testing conditions. They are usually obtained for a 99 wt% pure DCPO, analyzed using a high-pressure sealed pan to avoid evaporation during scanning. This means that the decomposition products will remain inside the capsule, interacting with the liquid DCPO (fusion point $\sim 50^\circ\text{C}$). Such conditions of decomposition may not be representative of the decomposition that takes place in the PSZ20/DCPO solution, where further reactions that do not occur in the experiment reported in [36, 37, 39–41] could also take place. This may explain that the values of heat of decomposition obtained in [36, 37, 39–41] differ of about 10 to 30% from the value

TABLE 2. Kamal-Sourour model parameters obtained with isothermal identification ($x = 1$ wt%) and n^{th} -order model parameters obtained with dynamic scans ($x = 5, 10$, and 20 wt%).

Kamal-Sourour model for polysilazane crosslinking			
$k_{o1} = 1.66 \times 10^{13} \text{ s}^{-1}$	$E_{a1} = 124.73 \text{ kJ}\cdot\text{mol}^{-1}$	$m = 2.54$	
$k_{o2} = 2.38 \times 10^{16} \text{ s}^{-1}$	$E_{a2} = 143.12 \text{ kJ}\cdot\text{mol}^{-1}$	$n = 1.11$	
n^{th} -order model for DCPO decomposition (compound)			
$k_d = 5.00 \cdot 10^{13} \text{ s}^{-1}$	$E_d = 130.00 \text{ kJ}\cdot\text{mol}^{-1}$	$n_d = 1.15$

FIG. 11. Model results for several DCPO concentrations compared to experiment: (a) $x = 0.1$, (b) 0.5, (c) 1, (d) 3, (e) 5, (f) 10, and (g) 20 wt%; (h) contribution of PSZ20 polymerization and DCPO decomposition to the overall heat of reaction computed for $x = 10$ wt%.

identified previously with Eq. 4. For this reason, the values of kinetics model parameters need to be identified from the present experiments performed using several compounds, even if the assumption of an n^{th} -order model will be considered to describe DCPO decomposition. The experimental results obtained using DCPO only have not been used to identify the parameters of Eq. 7.

PSZ20 polymerization will be modeled using a Kamal-Sourour phenomenological model [46]. Several authors used this

law to model epoxy resins polymerization [47–49] or silicone rubber vulcanization [50], because of the autocatalytic and catalytic nature of the reactions driving the crosslinking of these systems. A previous work [22] shows that this model also satisfactorily describes the kinetics behavior of the compound of PSZ20 with 1 wt% of DCPO. This model, slightly modified in Eq. 8, includes two rate constants that follow an Arrhenius temperature dependence. m and n are the reaction orders. The reactions whose kinetics is described by Eq. 8 are assumed to be

FIG. 12. Comparison of experimental and computed (a) peak temperature and (b) specific heat flow at peak for both first polymerization peak and second decomposition peak when it exists. The experimental data related to the second peak for $x = 5$ wt% are measured at the minimum slope, since there is no local maximum on this curve.

initiated by DCPO. An increase of initiator concentration acts as increasing the frequency of the interactions, and therefore increases the frequency factor and accelerates the reaction. This effect is described by the factor $(x/x_{\text{ref}})^w$ added in the Kamal-Sourour model, Eq. 8. It is noteworthy that multiplying only the catalytic term by the factor $(x/x_{\text{ref}})^w$ do not allow a proper fitting of the experimental results.

$$\dot{\alpha}_p = \left(\frac{x}{x_{\text{ref}}}\right)^w \cdot \left(k_{o1}e^{-\frac{E_{p1}}{RT}} + k_{o2}e^{-\frac{E_{p2}}{RT}}\alpha_p^m\right)(\alpha_{p\text{max}} - \alpha_p)^n \quad (8)$$

In Eq. 8, the term $\alpha_{p\text{max}}$ allows accounting for the partial polyaddition of the vinyl groups for low DCPO concentration and for the dilution effect as a function of x , according to Eq. 5. The overall heat of reaction is thus introduced as a function of x by this way. However this term is not sufficient to describe the shift of peak conversion toward lower temperature with the increase of DCPO concentration. For that reason, the reaction rate, Eq. 8, includes explicitly the dependence to DCPO concentration, x . In Eq. 8, x_{ref} represents the DCPO concentration previously used in [22] to identify the K-S parameters, i.e. 1 wt%. This approach was already used in another paper [51]. In the present work, an adjustable exponent w is necessary to properly fit the experimental curves over the entire DCPO concentration range.

The parameters of the PSZ20 polymerization model, Eq. 8, are identified from the isothermal DSC scans reported in [22] for a DCPO concentration of 1 wt%. The DCPO decomposition was neglected thanks to the very low concentration in initiator; the same procedure as described in [22] was thus used, but the polymerization heat of reaction identified previously in section “Heat of Reaction” is considered, accounting for a value of $\alpha_{p\text{max}} < 1$. The rate constants for the present Kamal-Sourour corrected model, Eq. 8, are presented in Fig. 10. The parameters obtained for the Kamal-Sourour model in this paper (Table 2) are slightly different from those obtained in [22], where the effect of the model of decomposition was not taken into account and the maximal degree of polymerization was set to 1. For this reason, in the previous work, the parameters were overestimated with regard to the present work.

The value of the adjustable parameter w is obtained on the basis of the different dynamic scans with various DCPO concentrations. The value of 0.59 is identified in such a way that the

temperature at the first peak matches the experiment up to $x = 5$ wt%, i.e., PSZ20 polymerization is preponderant in the vicinity of the first peak.

The n^{th} -order model parameters (Eq. 7) are identified using dynamic DSC scans in such a way that the model properly describes the end of the second peak for $x > 5$ wt%, where PSZ20 polymerization can be considered as complete. An activation energy of $130 \text{ kJ}\cdot\text{mol}^{-1}$ is obtained, in the range identified in [36, 37, 40, 41]. The value of the other parameters (Table 2) are of the same order of magnitude as those reported in [36, 37, 40, 41] (Table 1).

The parameters of this global phenomenological model, described by the set of Eqs. 5–8, identified using this procedure, are reported in Table 2. According to the parameters presented in Table 2, it is possible to plot the model results under dynamic conditions ($2 \text{ K}\cdot\text{min}^{-1}$) for different DCPO concentrations (Fig. 11). The plot for 1 wt% (Fig. 11c) corresponds to the K-S model identified using isothermal conditions and the model of DCPO decomposition, while the other plots are obtained changing the DCPO concentration (parameter x) in Eqs. 5–8.

$$q = \Delta H_p \cdot (1-x) \cdot \dot{\alpha}_p + \Delta H_d \cdot x \cdot \dot{\alpha}_d \quad (9)$$

In Fig. 11 is presented the modeled overall specific heat flow, q , Eq. 9, (computed using the MatLab[®] subroutine *ode15s* to integrate the set of Eqs. 6–9 and the experimental data (Fig. 11a–g). The contribution of each reaction (PSZ20 polymerization and DCPO decomposition) to the overall heat flow is plotted in Fig. 11h for a DCPO concentration of 10 wt% and shows that each peak is mainly related to one reaction. The model satisfactorily reproduces the shape of the first peak related to PSZ20 polymerization irrespective of the DCPO concentration, as well as the progressive occurrence of a second peak while increasing the DCPO concentration. A more quantitative comparison between model and experiment is carried out in Fig. 12, where the peak temperature is plotted against DCPO concentration (Fig. 12a). The discrepancy between the modeled and measured values of the temperature at the first peak is about 1 K, and reaches 5 K for the second peak. The specific heat flow at peak (Fig. 12b) is also almost perfectly reproduced by the model, excepted for $x = 3, 5$, and 10 wt%, where the value at the first peak is significantly overestimated up to 10% for $x = 5$ wt%.

Despite the satisfactory results provided by the model concerning the global shape of the curve and the effect of DCPO concentration on the characteristics of both polymerization and decomposition peaks, a significant mismatch between experiment and model is obtained between both peaks, where the model underestimates the specific heat flow for $x \geq 0.5$ wt%. This phenomenon is probably related to the decomposition of peroxide, which may not exactly follow an n^{th} -order model in the compound (cf. section “DCPO Decomposition”), and may also affect the end of polymerization. Indeed the main hypothesis of this model is that both kinetic models are completely uncoupled, neglecting the interaction between peroxide decomposition and polymerization. This model is probably too simplistic and a more mechanistic approach would be unavoidable in order to integrate polymerization dependence upon DCPO decomposition. This could be done introducing a catalytic term inside the Kamal-Sourour equation depending on the degree of conversion of DCPO decomposition. However the mechanisms ruling this phenomenon are not well understood and for this reason a more in-depth analysis should be performed, studying the decomposition/polymerization kinetics under isothermal conditions for a higher DCPO concentration (20 wt%): under these conditions the reciprocal influence of both phenomena could be better understood and a more detailed model could be introduced.

CONCLUSIONS

The polymerization reaction of a polysilazane system was studied in presence or not of DCPO. The reaction without addition of DCPO takes place above 280°C, driven by the reactions of transamination and polymerization of vinyl groups. To accelerate PSZ20 polymerization and reduce the peak temperature, addition of DCPO is necessary. In presence of this initiator, vinyl groups polyaddition takes place around 120–150°C, depending on DCPO concentration. The optimal specific heat of polymerization is obtained for a concentration of 1.52 wt% of DCPO. Above this value the evolution of the overall heat of reaction can be described using a linear law, with a specific heat of DCPO decomposition of 964.92 J·g⁻¹ and a specific heat of vinyl groups polyaddition of 231.87 J·g⁻¹. Under this optimal value, a partial polyaddition reaction is observed, with a specific heat of polymerization depending on DCPO concentration through an exponential law. This describes a low probability of polyaddition initiation for low DCPO concentration. According to this interpretation it is proposed to describe the evolution of the final degree of conversion for both reactions of decomposition and polymerization. Cure kinetics is described according to two phenomenological models: an n^{th} -order model for the decomposition of DCPO and a modified Kamal-Sourour model for the polymerization of PSZ20. A DCPO concentration dependent term is introduced in the Kamal-Sourour model to take into account the evolution of polymerization kinetics with DCPO concentration. This model includes 14 parameters: 4 parameters describe the change in overall specific heat of reaction with DCPO concentration; 6 parameters for the Kamal-Sourour model are identified from a complete characterization reported in [22] involving isothermal and dynamic DSC tests; 3 parameters for the n^{th} -order model to describe the second peak; and one parameter to match the first peak shift with changing

DCPO concentration. This model reproduces very satisfactorily the evolution of peak temperatures and intensities of both reactions, although a little mismatch between experimental data and the results of the model is observed between both peaks.

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NOMENCLATURE

α	Overall degree of conversion (compound), dimensionless.
$\dot{\alpha}_i$	Temporal derivative of α_i (reaction rate), s ⁻¹ .
α_1	Degree of conversion for the first peak (DCPO alone), dimensionless.
α_2	Degree of conversion for the second peak (DCPO alone), dimensionless.
α_d	Degree of DCPO degradation (compound), dimensionless.
α_{DCPO}	Degree of degradation (DCPO alone), dimensionless.
α_p	Degree of polymerization (compound), dimensionless.
ΔH_d	Specific heat of DCPO decomposition (compound), J·g ⁻¹ .
ΔH_{DCPO}	Total specific heat of decomposition (DCPO alone), J·g ⁻¹ .
ΔH_p	Specific heat of polymerization reactions (compound), J·g ⁻¹ .
ΔH_{tot}	Total specific heat of reaction (compound), J·g ⁻¹ .
ΔH_1	Specific heat of decomposition for the first peak (DCPO alone), J·g ⁻¹ .
ΔH_2	Specific heat of decomposition for the second peak (DCPO alone), J·g ⁻¹ .
$\zeta = \frac{\Delta H_2}{\Delta H_{\text{DCPO}}}$	Part of ΔH_{DCPO} attributed to the second peak, dimensionless.
A_0	Fitting parameter of $\Delta H_{\text{tot}}(x)$ when x tends toward 0, dimensionless.
DCPO	Symbol for dicumyl peroxide.
E_1	Activation energy for the first peak (DCPO alone), kJ·mol ⁻¹ .
E_2	Activation energy for the second peak (DCPO alone), kJ·mol ⁻¹ .
E_{a1}	Activation energy of polymerization (catalytic component, compound), kJ·mol ⁻¹ .
E_{a2}	Activation energy of polymerization (autocatalytic component, compound), kJ·mol ⁻¹ .
E_d	Activation energy of DCPO degradation (compound), kJ·mol ⁻¹ .
k_{01}	Pre-exponential factor for polymerization (catalytic component, compound), s ⁻¹ .
k_{02}	Pre-exponential factor for polymerization (autocatalytic component, compound), s ⁻¹ .
k_1	Pre-exponential factor for the first peak (DCPO alone), s ⁻¹ .

k_2	Pre-exponential factor for the second peak (DCPO alone), s^{-1} .
k_d	Pre-exponential factor for DCPO degradation (compound), s^{-1} .
m	Pseudo-order of polymerization (autocatalytic component, compound), dimensionless.
n	Pseudo-order of polymerization (catalytic component, compound), dimensionless.
n_1	Pseudo-order of reaction for the first peak (DCPO alone), dimensionless.
n_2	Pseudo-order of reaction for the second peak (DCPO alone), dimensionless.
n_d	Pseudo-order of reaction for DCPO degradation (compound), dimensionless.
PSZ20	Commercial designation of the polysilazane resin.
q	Specific heat flow measured by DSC, $W \cdot g^{-1}$.
R	Ideal gas constant, $J \cdot mol^{-1} \cdot K^{-1}$.
T	Temperature, K if not specified.
x	DCPO weight fraction in the compound, dimensionless.
x_0	Fitting parameter of $\Delta H_{tot}(x)$ for $0 < x < 0.015$, dimensionless.
x_{ref}	DCPO weight fraction used to calibrate the Kamal-Sourour model from isothermal DSC tests in [22], dimensionless.
w	Fitting parameter that controls the peak temperature shift with varying x , dimensionless.

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